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Preparation of TiO₂ thin films by pulsed laser deposition for waveguiding applications

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Abstract

TiO₂ thin films have been deposited on SiO₂/Si, MgO and Al₂O₃ substrates by Pulsed Laser Deposition with a KrF excimer laser from a rutile ceramic target. From X-ray diffraction and Raman spectroscopy, their crystalline structure appear to be mainly anatase. The potentialities of waveguiding properties have been evaluated by measuring refractive index and thickness with m-line spectroscopy.

1. Introduction

TiO₂ is an important material for optical applications (as antireflective coating for example) and TiO₂ thin films have been grown by a number of techniques, each one of them emphasizing different advantages [1]. Surprisingly there are only few reports on Pulsed Laser Deposition (PLD) of TiO₂ [2-4], though the interest of this technique has been well recognized. If optical waveguide fabrication and characterization are extensively studied for integrated optics applications, PLD technique has only very recently been used, and only for nonlinear optical materials such as LiNbO₃ [5,6]. The aim of this study is to show the feasibility of growing by PLD TiO₂ thin films suitable for optical waveguiding applications.

2. Thin films deposition

The experimental set-up has already been described [7]. We give only the main parameters of the deposition procedure. The beam of a KrF excimer laser ($\lambda = 248$ nm) was focussed on the rotating target with a fluence of 3 J/cm². The temperature of the substrate, heated by an halogen lamp, was 790°C. The base pressure in the deposition chamber was 4×10^{-8} mbar but the film deposition took place in an oxygen pressure of 10^{-1} mbar. The cooling of the films was achieved in a 200 mbar pressure of oxygen. The targets have been prepared from TiO₂ rutile powder with very fine grains (around 10 nm), compressed and sintered at 1050°C in air up to 56% of the bulk density, keeping the rutile structure as checked by X-ray diffraction (XRD) and Raman spectroscopy (see Section 3.2). Films have been grown on carefully cleaned monocrystalline substrates: MgO (100), Si covered with a SiO₂ amorphous layer, about 1 μ m thick, and Al₂O₃. The discussion will concentrate more specifically on these samples.

The thickness of the films was determined with a Sloan Dektak IIA profilometer. Films of less than 1 μ m thickness have been obtained after a 40 mn deposition. Optical microscopy observations show that the samples have good transparency and are homogeneous, crack-free, with no droplets. Some films have been grown under other conditions (Table I) and the influence of the deposition parameters will be discussed briefly in the next section.

3. Structural characterization

TiO₂ exists in three different crystalline phases: anatase, rutile (both tetragonal) and brookite (orthorhombic). Amorphous TiO₂ crystallises upon heating into anatase, which transforms into rutile above about 700°C. Depending on the deposition techniques and parameters, it is possible to obtain amorphous TiO₂ (for low substrate temperature generally) or the anatase and rutile phases or both, with various proportions, orientations and crystal sizes [1,8]. For the structural characterization we used X-ray diffraction and Raman spectroscopy.

3.1. X-ray diffraction

Fig. 1 shows the XRD diagrams of TiO₂ thin films deposited on SiO₂/Si, and MgO substrates, registered with a Siemens D500 diffractometer (λ Cu K α = 1.540 Å, graphite back-monochromator). In the case of SiO₂/Si substrate the spectrum contains essentially the (004) line of anatase with very weak (101) and (200) lines. The film is thus constituted of partially oriented anatase crystals, the preferential orientation of the *c*-axis being perpendicular to the film surface. A small amount of rutile is also detected, as some weak rutile lines, (101), (111) and (211) are observed.

For the film deposited on the oriented MgO substrate, besides the very intense line due to the (200) plane of MgO, the same (004) anatase line is predominant in the spectrum. The same preferential orientation of the *c*-

axis of the anatase crystals, perpendicular to the film surface, is obtained. In addition weak (112) and (200) lines are observed for anatase. The contribution of rutile, detected by the (200) and (111) lines, is weak.

3.2. Raman spectroscopy

The films were investigated using a Dilor X-Y modular multichannel laser Raman spectrometer equipped with a confocal microscope (backscattering configuration). The exciting wavelength is 514.5 nm. With the used objective (X 50), the analysed area has a diameter of about 10 μm .

Fig. 2 compares the Raman spectra of the rutile target and the film deposited on SiO_2/Si . The target exhibits the characteristic rutile spectrum in this range [9], with bands at 612 cm^{-1} (A_{1g}), 447 cm^{-1} (E_g), and 143 cm^{-1} (B_{1g}). The 235 and 360 cm^{-1} lines are attributed to multiphonon scattering. The spectrum of the film, besides the 521.5 cm^{-1} line due to the Si substrate, is mainly constituted by the anatase spectrum [10]: 639 cm^{-1} (E_g), 516 cm^{-1} ($A_{1g}-B_{1g}$), 399 cm^{-1} (B_{1g}), 197 cm^{-1} (E_g), 144 cm^{-1} (E_g). Two weak bands at 618 and 442 cm^{-1} show that some amount of rutile phase is also present. Changing the polarization of the exciting laser beam does not modify significantly the relative line intensities. This is in agreement with a preferential orientation of the *c*-axis perpendicular to the film surface, taking into account the polarization selection rules for the Raman spectrum of anatase [10].

Fig. 3 gives the Raman spectra of the film grown on a MgO substrate for two polarizations. The anatase spectra are recognized, with a weak rutile contribution. The change in the relative intensities of the lines with polarization indicates that the *c*-axis preferential orientation perpendicular to the film surface is less well defined than for the SiO_2/Si substrate. All these results are in agreement with those of X-ray diffraction. However further information may be obtained, as will be discussed below.

4. Discussion

It is interesting to notice first that, under these deposition conditions (0.1 mbar of O_2 , 790°C as substrate temperature), the predominant phase is anatase, though its growth takes place at a temperature above that at which it is supposed to begin to transform into rutile (700°C). However, if rutile is the thermodynamically stable phase, anatase may be kinetically stable until about 1000°C , depending on the grain size for example. Such a situation is similar to that of sol-gel thin films where anatase alone has been observed even after annealing at 900°C [11].

Another point of interest is that anatase has a preferential orientation of the *c*-axis perpendicular to the film surface, for both a monocrystalline substrate, MgO, and an amorphous one, SiO_2 , deposited on Si. The film deposited on Al_2O_3 is amorphous according to XRD, but shows some of the rutile and anatase lines in the Raman spectrum. As the coherent domain sizes for the two processes should be different, it may be inferred that very small and/or imperfect crystallites are present.

For films #1 and #2 the width of the anatase main line at 144 cm^{-1} is broaden relative to that of the bulk crystal (14 cm^{-1} instead of 7 cm^{-1}) [12]. The line width can be related to the mean size of the crystals, which could be of the order of 10 nm as it has been observed for anatase powders [13] and in anatase thin films deposited by the sol-gel method [11].

Our results differ from previous reports on TiO_2 deposition on Si wafers, as could be expected for different deposition parameters. Depositions with a YAG laser at 532 nm under vacuum lead to polycrystalline rutile films for a substrate temperature of 850°C , with a few anatase phase at 500°C . Films grown at lower substrate temperatures are amorphous [2], as those deposited with an ArF laser at substrate temperature less than 200°C , under various O_2 pressures [3]. At last, for a deposition with an ArF laser at 450°C without introduction of O_2 , rutile is obtained with a (200) orientation and when increasing the O_2 pressure from 0.1 T to 0.3 T, one goes from rutile (with (110) and (200) lines) to anatase (with (101) and (004) main lines) [4].

Preliminary results on the influence of deposition parameters for a MgO substrate (Table 1) show that rutile is preponderant at higher laser fluence, higher oxygen pressure and if a rutile ceramic target prepared with coarser grains is used. Better understanding of the deposition process could result from the study of the nature and energy of the species present in the plume [14].

4.1. Optical characterization

The optical waveguiding potentiality has been tested by the dark m-line spectroscopy, the principle of which is briefly recalled. The beam of an He-Ne laser ($\lambda = 632.8\text{ nm}$) is coupled into the film with the aid of a rutile prism (angle: 45.38°): for discrete values of the incidence angle, the evanescent wave, existing in the air gap between the prism and the film, may excite the guided modes in the film. The coupling is detected by the apparition of dark lines in the reflected beam and the effective index may be calculated for each mode [15].

For sample #1 (TiO_2 on SiO_2/Si) five TE (transverse electric) modes and only three TM (transverse magnetic) modes are observed due to a low coupling efficiency. Assuming a step index profile, the thickness (930 nm) and the refractive index ($n = 2.219$) have been deduced from the TE modes. For sample #2 (TiO_2 on MgO) four TE and four TM modes are pointed, but, as for sample #1, the dark lines are large (up to 0.5°), inducing a low precision of the measurements. Both TE and TM observations provide a thickness of $870 \pm 3\text{ nm}$.

The refractive index are ranging from 2.23 (TE) to 2.25 (TM) which is in agreement with an anisotropy of the layer observed by XRD and Raman spectroscopy. Though light coupling into the film could be achieved with a prism, it was not possible to detect any propagation on a significant distance. This can be due to several reasons: surface roughness, porosity at the scale of the wavelength, as the refractive index is smaller than for the bulk crystals ($n_0 = 2.605$, $n_e = 2.899$ for rutile, $n_0 = 2.488$, $n_e = 2.561$ for anatase, at 589.3 nm).

For the film #4 (TiO₂ on MgO deposited at lower temperature) two dark lines, very sharp (less than 0.05 degree), are observed for the TE, and TM, modes. As this film is thin (less than 200 nm) due to a shorter deposition time, only the first two modes exist, which does not allow the calculation of the thickness and refractive index. The good quality of this film is attested by the sharpness of the lines. In this case a good hope exists of observing guided wave propagation. Unfortunately, due to the small size of the sample (5 X 5 mm²), this could not be checked. At last, waveguiding over a distance of about 1 cm was observed in TiO₂ film #3 on Al₂O₃.

5. Conclusion

We have grown TiO, thin films on MgO, SiO₂/Si and Al₂O₃ substrates by PLD from rutile ceramic targets. The films are mainly constituted by small crystals of anatase, r-rutile or both, depending on the deposition parameters. These crystals are partially oriented, as characterized by X-ray diffraction and micro-Raman spectroscopy. We measured their thickness and refractive index by m-line spectroscopy. Favorable waveguiding properties may be expected, depending on substrate and film structure.

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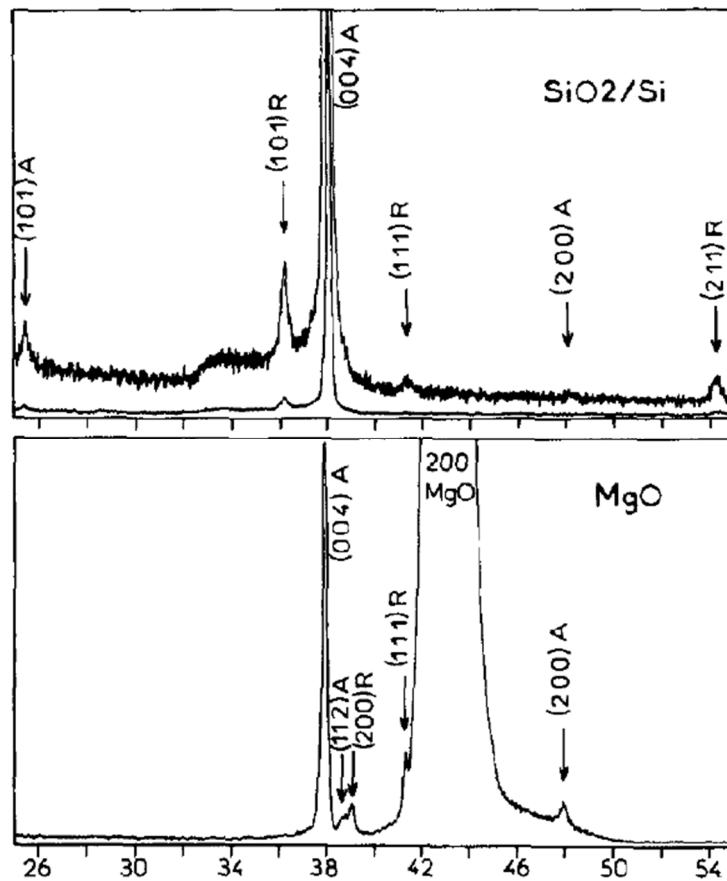


Fig. 1. X-ray diffraction diagrams of TiO_2 films deposited on SiO_2/Si and MgO

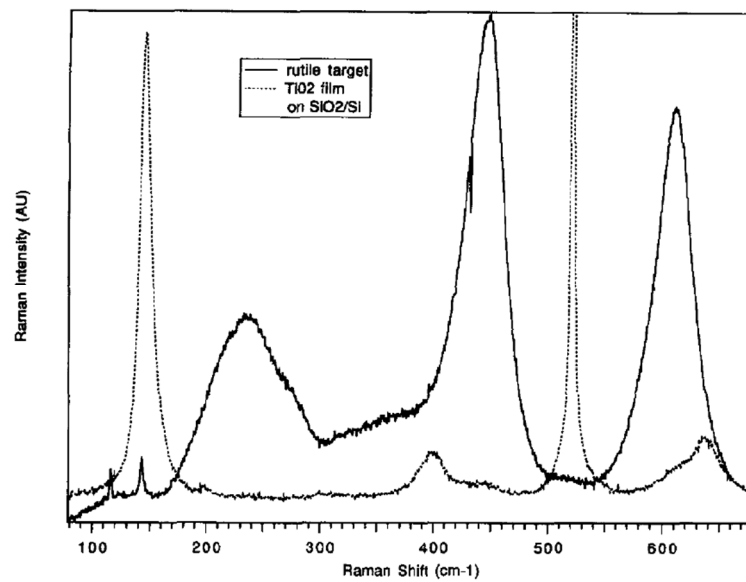


Fig. 2. Raman spectra of the rutile target and TiO_2 films on SiO_2/Si

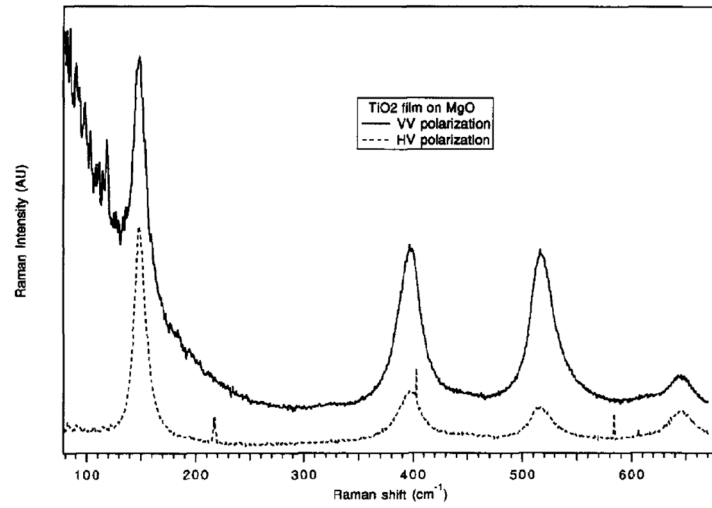


Fig. 3. Raman spectra of TiO₂ film #1 deposited on MgO.

Table 1 : Deposition parameters of the samples and predominant crystalline phase observed by X-ray diffraction and Raman spectroscopy

Sample	Substrate	Substrate temperature (°C)	Oxygen pressure (mbar)	Ceramic target	Laser fluence (J/cm ²)	Predominant crystalline phase ^a	
						Rx	Raman
1	MgO(100)	790	10 ⁻¹	rutile	3	A	A
2	SiO ₂ /Si	790	10 ⁻¹	rutile	3	A	A
3	Al ₂ O ₃	790	10 ⁻¹	rutile	3	am	A + R
4	MgO	600	3 × 10 ⁻¹	rutile	3	R	—
5	MgO	790	3 × 10 ⁻¹	rutile	3	R	—
6	MgO	790	10 ⁻¹	rutile ^b	3	R	R
7	MgO	790	10 ⁻¹	rutile	6	R	R

^a R: rutile, A: anatase, am: amorphous.

^b Coarse grains.